

Titlu proiect: *Influence of lead doped manganites of type $La(Ho,Nd)_{0.54}Sr(Ca,Ba)_{0.35-x}Pb_xMnO_3$ on the microstructural, magnetic and transport properties*

Categoria de proiect: proiect de cercetare

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Manager proiect: Conf. Dr. Nicoleta Cornei

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| 1 | M.-L. Craus, V. Dobra, E. Anitas, R. Erhan, N. Cornei, | <i>Magnetic structure and transport phenomena in Pb low doped $LaMnO_3$ manganites</i> | The 11 th edition of the EMSA Conference Torino, Italy, EU | 12-15 iulie 2016 |
| 2 | M.-L.Craus, E. M. Anitas, N. Cornei, V. Turchenko | <i>Magnetic Structure and Transport Phenomena in $La_{0.5}Pr_{0.2}Pb_{1-x}Sr_xMnO_3$ manganites</i> | The 11 th edition of the EMSA Conference Torino, Italy, EU | 12-15 iulie 2016 |
| 3 | N. Cornei, C. Mîță, M.-L. Craus, I. A. Gorodea, V. Dobra, A. Domocoș | <i>Synthesis, Structural and Catalytic Properties of $LaFe_{1-x}Co_xO_3$ perovskites.</i> | 16 th International Balkan Workshop on Applied Physics and Materials Science - IBWAP 2016 | 7-9 iulie 2016 |

Lista achizițiilor realizate în cadrul proiectului

| Denumire echipament | Categorie de achiziții: (numai mijloace fixe) | Valoarea (lei) | Locație |
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Director proiect,
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The 11th edition of the EMSA Conference Torino, Italy, EU, July 12 through July 15, 2016.



Magnetic structure and transport phenomena in Pb low doped LaMnO₃ manganites

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The aim of this paper is to establish the influence of Pb substitution on magnetic/crystalline structure and transport properties of LaMnO₃ manganites

EXPERIMENTAL DETAILS

Ceramic technology: The La_{1-x}Pb_xMnO₃ manganites were synthesized by ceramic technology at Laboratory of Neutron Physics, JINR, Dubna, Russia. The samples were finally treated at 1200°C in a closed vessel to prevent Pb evaporation.

X-ray Diffraction (XRD): X-ray diffraction (XRD) experiments were performed at room temperature in order to determine the phase composition and microstructural parameters by using Fullprof software. The space group and lattice constants were obtained by means of the CeckCell and proofed by FullProf software.

Magnetic measurements: Foner type magnetometer, between 77 and 400K.

Electric measurements: Four points method. Resistance measurements with temperature and magnetic field intensities were performed by using a closed cycle refrigerator and an electromagnet at JINR, Dubna, Russia.

RESULTS

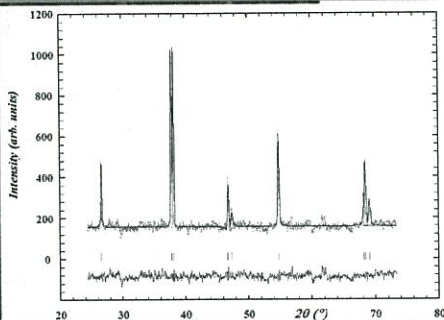


Figure 1. Observed (black) and calculated (red) diffractograms of La_{0.94}Pb_{0.06}MnO₃ using FullProf software; bottom (green) – the difference between the calculated and observed diffractograms

Table 1. Variation of lattice constants (a, b, c), unit cell volume (V_{uc}) and occupied by a molecule (V_{uc}/Z) for La_{1-x}Pb_xMnO₃ manganites volume

| x | a=b (Å) | c (Å) | V _{uc} (Å ³) |
|------|---------|---------|-----------------------------------|
| 0.03 | 5.5320 | 13.3778 | 354.6 |
| 0.06 | 5.5272 | 13.3625 | 353.5 |
| 0.10 | 5.5273 | 13.3667 | 353.7 |
| 0.15 | 5.5297 | 13.3703 | 354.1 |
| 0.20 | 5.5256 | 13.3628 | 353.3 |

Structure The synthesized manganites contain only the perovskite phase: SG R-3c, with La or Pb on (6a) sites (0, 0, 0.25), Mn on (6b) sites (0, 0, 0) and O on (18e) (X, 0, 0.25), for x = 0.00, 0.05, 0.10, 0.15 and x = 0.20. The cell parameters, average size of crystalline blocks and microstrains have been obtained through the profile matching stage of the Rietveld refinement (Fig. 1 and Tab. 1). Only a small variation of the unit cell volume was observed with the increase of the Pb concentration and we attribute this behaviour to the presence of a small concentration of Pb⁴⁺ and/or to a small amount of the vacancies on both type of sites (Tabs.1 and 2).

Table 2. Variation of the average size of mosaic blocks (D), microstrains (ε), space group (SG) and calculated tolerance factor (t) for La_{1-x}Pb_xMnO₃ manganites.

| x | D (Å) | ε | t |
|------|-------|---------|-------|
| 0.03 | 1275 | 0.00006 | 0.967 |
| 0.06 | 1080 | 0.00031 | 1.000 |
| 0.10 | 1230 | 0.00014 | 1.000 |
| 0.15 | 1176 | 0.00005 | 0.995 |
| 0.20 | 1028 | 0.00001 | 0.995 |

Magnetic properties We suppose that a part of the Mn³⁺ cations does not participate to the total magnetic moment or they are forming an antiferromagnetic compound, with the same chemical composition and the same structure as the main phase. The substitution of La with Pb, together with the oxygen non-stoichiometry contributes to the appearance of the ferromagnetic interactions, and implicitly to a metallic phase. The variation of magnetization vs. temperature at relatively small intensities magnetic fields is similar to that observed in the samples characterized by the spin glass state, which were cooled in magnetic field (Figs. 2, 3).

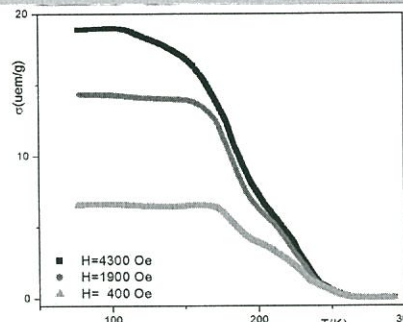


Figure 2 Variation of specific magnetization vs temperature and Pb concentration of LPMO (x=0.06) manganites, magnetic field cooled.

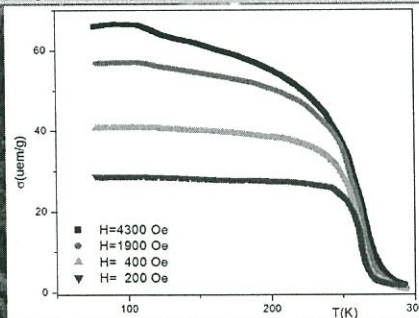


Figure 3 Variation of specific magnetization vs temperature and magnetic field intensity of La_{1-x}Pb_xMnO₃.

For all samples a spin glass transition was observed at temperatures apparently higher as 100 K. Examples were given in Figs. 2 and 3. The reasons for this behaviour is the decrease of the Pb⁴⁺ concentration and of the oxygen concentration, in the samples, with the increase of Pb concentration.

Transport characteristics The transport measurements were performed on field cooled and without field cooled samples, only on heating, with and without applied magnetic field (Figs. 4 and 5).

At low temperature the variation of the resistance can be described with relation:

$$R(T) = R_{4.5}T^{4.5} + R_2T^2 + R_0$$

where the last term (R₀) represents the resistance due to the temperature-independent scattering processes, the second term (in T²) corresponds to the electron–electron scattering processes, and the first term (in T^{4.5}) includes contribution from the electron–magnon scattering or spin wave scattering. These terms are more or less field dependent.

The larger values of resistance for the sample corresponding to x=0.06 is probably due to the appearance of a (nano)magnetic phase with a large resistivity, possibly as magnetic clusters based on Mn³⁺ or Mn⁴⁺ (s. Figs. 2 and 4).

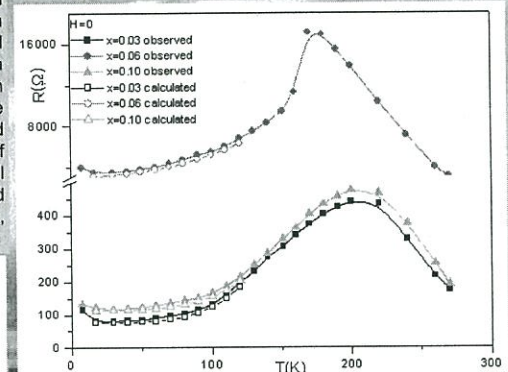


Figure 4 Variation of resistance vs temperature and Pb concentration, without magnetic field cooled; measurements were performed at H=0. For low temperature we observed a good agreement between the calculated and observed data.

CONCLUSIONS

1. We synthesized a set of pure La_{1-x}Pb_xMnO₃ manganites for various Pb concentrations (x = 0.03, 0.06, 0.10, 0.15 and 0.20).
2. All the samples have a rhombohedral structure (R-3c), with an almost constant unit cell volume. We observed a small variation of the lattice constants, despite the difference between the radii of La³⁺ (1.216 Å) and Pb²⁺ (1.35 Å), probably due to a change of oxygen concentration or/and an increase of Pb⁴⁺ concentration with total Pb concentration in the samples.
3. We have shown that they exhibit a spin glass state to metallic state transition at temperatures lower than 170 K.
4. There is a qualitative dependence between the presence of spin glass state and the observed large magnetoresistance. We intend to perform more exact experiments concerning the dependence between the presence (concentration) of spin glass state and the variation of resistance at low temperature.

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Magnetic structure and transport phenomena in Pb low doped LaMnO₃ manganites

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The La_{1-x}Pb_xMnO₃ (x=0.03 - 0.10) manganites were obtained by ceramic technology, from oxides, mixed in corresponding ratio. The sintering was performed at 1100°C, in a closed quartz tube, to prevent Pb evaporation. X-ray diffraction data at room temperature indicated that the samples contain only a distorted R-3c rhombohedral phase (s. Fig. 1). The cell parameters, average size of crystalline blocks and microstrains have been obtained through the profile matching stage of the Rietveld refinement (s. Fig. 1). We observed a small variation of the lattice constants, despite the difference between the radii of La³⁺ (1.216 Å) and Pb²⁺ (1.35 Å), probably due to a change of oxygen concentration or/and an increase of Pb⁴⁺ concentration with total Pb concentration in the samples. The oxygen parameter (X), d_{Mn-O}(Å), d_{A-O}(Å) distances and ∠Mn-O-Mn(°) angles were determined by using Fullprof code.

Curie temperature and specific magnetization at low temperature, determined with a VSM remain practically independent on Pb concentration.

At low temperature the variation of the resistance can be described with relation [1]:

$$R(T) = R_{4,5}T^{4.5} + R_2T^2 + R_0 \quad (1)$$

where R₀ represents the resistance due to the temperature-independent scattering processes, R₂ corresponds to the electron–electron scattering processes, and R_{4,5} includes contribution from the electron–magnon scattering or spin wave scattering. The larger values of resistance for the sample corresponding to x=0.06 is probably due to the appearance of a (nano)magnetic phase with a large resistivity, possibly as magnetic clusters based on Mn³⁺ or Mn⁴⁺ (s. Fig.2).

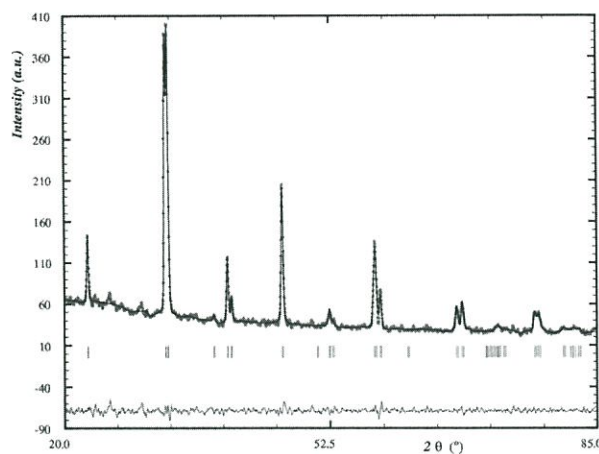


Figure 1 Diffractograms of La_{0.97}Pb_{0.03}MnO₃: at zero intensity - Bragg position of calculated maxima (vertical segment) and the difference between observed and calculated diffractograms, by using FullProf code.

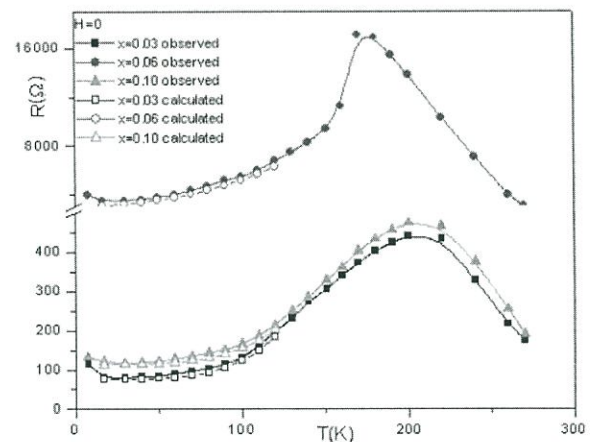


Figure 2 Variation of resistance with temperature for La_{1-x}Pb_xMnO₃ (H=0)

SYNTHESIS, STRUCTURAL AND CATALYTIC PROPERTIES OF $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ PEROVSKITESNicoleta CORNEI¹, Carmen MÎȚĂ¹, Mihail-Liviu CRAUS^{2,3}, Ioana Aurelia GORODEA¹, Viorel DOBREA² and Andrei DOMOCOȘ⁴¹University of „Al. I. Cuza” Iasi, Faculty of Chemistry, Bd. Carol I, 700506, Romania²National Institute of Research and Development for Technical Physics, Iasi, Romania;³Joint Institute of Nuclear Research, Dubna, Russia;⁴Faculty of Physics, “Alexandru Ioan Cuza” University, 11 Carol I Bvd., 700506 Iasi, Romania;

PURPOSE

The purpose of the present work is to obtain powders of $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ perovskite by sol-gel method, with a good catalytic activity.

INTRODUCTION

> The perovskites oxides contain more than one metallic element in their lattice. Due to the different amount of metals, the perovskites play an important role as catalysts in various chemical reactions [1, 2]. It can be used in many reactions due to its excellent redox properties, high oxygen mobility and thermal stability [3,4].

> By replacement of B cations with cations of similar size but lower oxidation state can be generate the vacancies.

> The main application of perovskite as catalysts has been in the oxidation of hydrocarbons and carbon monoxide.

> Recently, the perovskites oxides have been studied in the Fischer-Tropsch synthesis [5, 6].

EXPERIMENTAL DETAILS

Sol-gel method:

> The mixed oxide $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ ($x = 0; 0.05; 0.1; 0.2; 0.3$) was prepared by sol-gel self-combustion method using nitrates as precursors and citric acid and ethylene glycol as chelating agent. The obtained powder was annealed at 750 °C for 6 hours in air atmosphere.

> X-ray Diffraction (XRD): was performed with a Shimadzu XRD 6000 at room temperature (Cu K α 1 radiation and the step 0.02). Lattice constants and space group were determined and refined by using Powder cell and Celref programs.

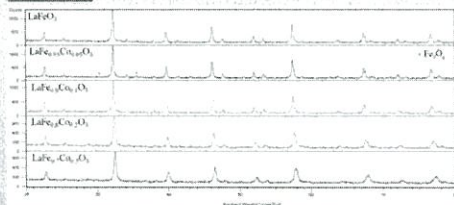
> FT-IR analysis: JASCO 660 Plus spectrophotometer in wave number range 4000 cm^{-1} – 350 cm^{-1} , in atmospheric air, in KBr disks

> BET analysis: Nova 2200e BET analyzer

> The H_2O_2 catalytic decomposition experiments were conducted in a batch reactor that was stirred at a speed to provide a uniform distribution of mixed oxide catalyst. The pH was adjusted by using a Na_2HPO_4 buffer solution. In the reaction vessel 2.5 mL H_2O_2 (0.05 M), 5 mL buffer solution, 7.5 mL distilled water and 0.05 g catalyst were added. The temperature of reaction chamber was maintained at 25 °C (± 0.1 °C) by using a Haake F4 pump. Oxygen evolution measurements were carried out with a water barometric system at normal pressure. All experiments were performed under the same conditions.

RESULTS AND DISCUSSION

XRD data:

Figure 1. X-ray diffraction for $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ perovskites annealed at 750 °C/6h

> The samples $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ became almost pure after a treatment for 6 hours at 750 °C (figs. 1 and 2).

> Structural phase identification was done using program's database X'Pert Plus High Score.

> The samples with $x = 0.05$ and 0.2 contain a small quantity of Fe_2O_3 (98-007-8086) and Co_3O_4 (98-002-6743)

> The results of X-ray diffraction indicated that $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ had a orthorhombic structure (space group: Pbnm, according with crystallographic data (fig.2).

> The cell parameters, average size of mosaic blocks and microstrains, have been obtained through the profile matching stage of the Rietveld refinement (tab. 1).

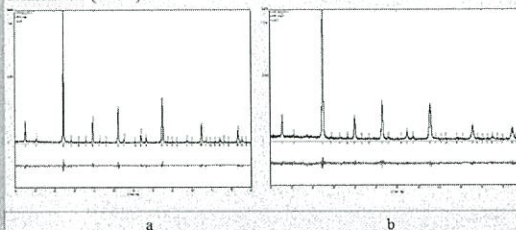
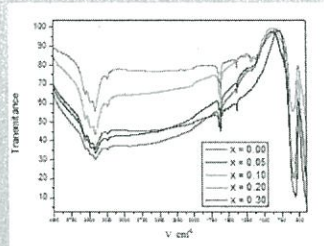
Figure 2. XRD powders of $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ (calculated and observed), annealed at 750 °C: a) $x = 0.0$; b) $x = 0.3$

Table 1. The variation of cell parameters (a, b, c), unit cell volume (V), average size of mosaic blocks (D), microstrains (ϵ) for $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ perovskites and BET surface area.

| Chemical composition | a(Å) | b(Å) | c(Å) | V(Å ³) | D(Å) | ϵ | BET surface area (m ² /g) |
|--|--------|--------|--------|--------------------|------|------------|--------------------------------------|
| LaFeO_3 | 5.5721 | 5.5537 | 7.8550 | 243.08 | 1075 | 0.000228 | 1.418 |
| $\text{LaFe}_{0.95}\text{Co}_{0.05}\text{O}_3$ | 5.5686 | 5.5520 | 7.8531 | 242.79 | 1031 | 0.000247 | 1.285 |
| $\text{LaFe}_{0.9}\text{Co}_{0.1}\text{O}_3$ | 5.5661 | 5.5504 | 7.8486 | 242.48 | 856 | 0.000122 | 2.179 |
| $\text{LaFe}_{0.8}\text{Co}_{0.2}\text{O}_3$ | 5.5216 | 5.5468 | 7.8431 | 240.21 | 655 | 0.000910 | 1.227 |
| $\text{LaFe}_{0.7}\text{Co}_{0.3}\text{O}_3$ | 5.5171 | 5.5242 | 7.8189 | 238.30 | 764 | 0.002235 | 1.583 |

> The D values decrease from LaFeO_3 to $\text{LaFe}_{0.7}\text{Co}_{0.3}\text{O}_3$ powders, that indicates that the doping with element cobalt in LaFeO_3 may prevent the grain growth during high temperature treatment.

IR data:

Figure 3. FTIR spectra of samples $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ perovskites annealed at 750 °C

> The spectrum shows bands at 1640 and 1616 cm^{-1} assigned to the asymmetric and symmetric O-C-O stretching vibration modes of acid carbonate associated with the presence of the CO_2 and H_2O adsorbed on the oxide surface.

> The broad band in 3550-3300 cm^{-1} range is associated to the water's O-H bond due to exposure to air and environmental humidity for the studied perovskites.

> IR absorption spectra of all samples, in the 400-700 cm^{-1} show two intense absorption bands assigned to symmetric and asymmetric vibrations deformation of the Co-O-Co/Fe-O-Fe bonds, whose frequencies are centered at 603 ± 3 cm^{-1} and 556 ± 2 cm^{-1}

BET analysis

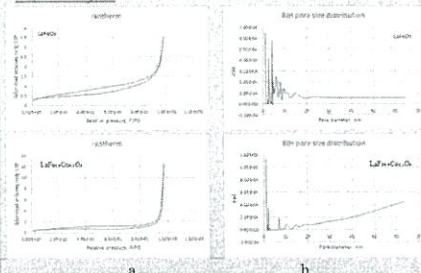


Figure 4. BET analysis: a) Isotherm of adsorption; b) Pore size distribution

> From BET isotherm we observe a variation of III type, specific for nonporous samples. Because they present a hysteresis loop, we can say that the samples have a poor porosity.

> BET surface area is in table 1.

Catalytic Activity

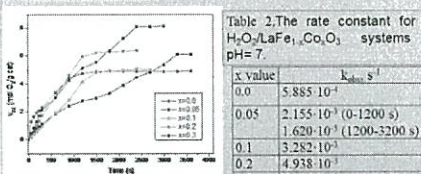
Figure 5. Variation of O_2 quantity with time in presence of $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ as catalyst

Table 2. The rate constant for the $\text{H}_2\text{O}_2/\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$ systems pH=7.

| x value | k_{obs} , s ⁻¹ |
|---------|-------------------------------------|
| 0.0 | $5.885 \cdot 10^{-4}$ |
| 0.05 | $2.155 \cdot 10^{-4}$ (0-1200 s) |
| 0.1 | $1.620 \cdot 10^{-4}$ (1200-3200 s) |
| 0.2 | $3.282 \cdot 10^{-4}$ |
| 0.3 | $7.120 \cdot 10^{-4}$ (0-400 s) |
| | $2.713 \cdot 10^{-4}$ (400-2400s) |

> The decomposition of oxygen peroxide: $\text{H}_2\text{O}_2 = \text{H}_2\text{O} + 1/2\text{O}_2$ follows a pseudo-first kinetic order rate law $\log[\text{H}_2\text{O}_2] = \log[\text{H}_2\text{O}_2]_0 - k_{\text{obs}}t/2.3$ at square correlation coefficient (R^2) > 0.980, where $[\text{H}_2\text{O}_2]_0$ and $[\text{H}_2\text{O}_2]$ are the concentrations of H_2O_2 in solution time zero and any time t, k_{obs} is observed rate law.

> The k_{obs} values and kinetic relationship indicates that the rate determining step in H_2O_2 decomposition could be diffusion of chemicals to the surface, adsorption on the reactive sites, electron transfer, desorption of products or regeneration of the reactive sites

CONCLUSION

> The samples present an orthorhombic structure
 > The presence of CO_2 and H_2O adsorbed on the oxide surface suggest that perovskites have hydrophilic properties and good catalytic activity
 > The obtained samples have potential application as gas sensor

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The aim of this paper is to establish the influence of Pb substitution with Sr and of La with Pr on magnetic/crystalline structure and transport properties of $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.3-x}\text{Sr}_x\text{MnO}_3$ manganites.

EXPERIMENTAL DETAILS

Ceramic technology: The $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.3-x}\text{Sr}_x\text{MnO}_3$ manganites were synthesized by ceramic technology at Laboratory of Neutron Physics, JINR, Dubna, Russia. The samples were finally treated at 1200°C in a closed vessel to prevent Pb evaporation.

X-ray Diffraction (XRD): X-ray diffraction (XRD) experiments were performed at room temperature in order to determine the phase composition and microstructural parameters by using Fullprof software. The space group and lattice constants were obtained by means of the CeckCell and profed by FullProf software.

Magnetic measurements: Foner type magnetometer, between 77 and 400K.

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RESULTS

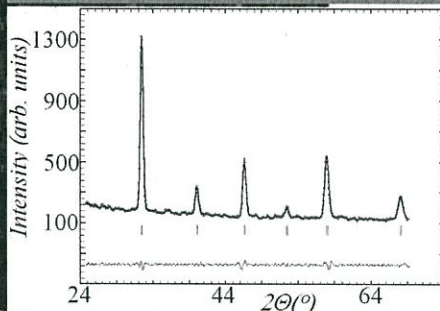


Figure 1. Observed (black) and calculated (red) diffractograms of $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.3}\text{MnO}_3$ using FullProf software; bottom (blue) – the difference between the calculated and observed diffractograms

Table 1. Variation of lattice constants (a, b, c), unit cell volume (V_{uc}) and occupied by a molecule (V_{uc}/z) for $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.3-x}\text{Sr}_x\text{MnO}_3$ manganites volume.

| x | a=b (Å) | c (Å) | V_{uc} (Å ³) | V_{uc}/z (Å ³) |
|------|---------|--------|----------------------------|------------------------------|
| 0.00 | 3.880 | 3.880 | 58.44 | 58.44 |
| 0.05 | 3.896 | 3.896 | 58.86 | 58.86 |
| 0.10 | 5.540 | 13.419 | 356.66 | 59.44 |
| 0.15 | 5.513 | 13.355 | 350.36 | 58.39 |
| 0.20 | 5.524 | 13.385 | 353.70 | 58.95 |

Structure The synthesized manganites contain only the perovskite phase: 1) SG Pm-3m with La/Pr/Sr or Pb on (1b) sites (1/2, 1/2, 1/2), Mn on (1a) sites (0, 0, 0) and O on (3d) sites (1/2, 1/2, 0) for x = 0 and 0.05 or 2) SG R-3c, with La/Pr/Sr or Pb on (6a) sites (0, 0, 0.25), Mn on (6b) sites (0, 0, 0) and O on (18e) (X, 0, 0.25), for x = 0.10, 0.15 and x = 0.20. The cell parameters, average size of crystalline blocks and microstrains have been obtained through the profile matching stage of the Rietveld refinement (Fig. 1 and Tab. 1). We observed an increase of unit cell volume for each of the two types of structures separately and we attribute this behaviour to the presence of a small concentration of Pb^{4+} and/or to a small amount of the vacancies on both type of sites (Tabs.1 and 2).

Table 2. Variation of the average size of mosaic blocks (D), microstrains (ϵ), space group (SG) and calculated tolerance factor (t) for $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.3-x}\text{Sr}_x\text{MnO}_3$ manganites.

| x | D (Å) | ϵ | SG | t |
|------|-------|------------|-------|-------|
| 0.00 | 315 | 0.0010 | Pm-3m | 1.000 |
| 0.05 | 551 | 0.0028 | Pm-3m | 1.000 |
| 0.10 | 845 | 0.0015 | R-3c | 0.963 |
| 0.15 | 600 | 0.0005 | R-3c | 0.995 |
| 0.20 | 480 | 0.0007 | R-3c | 0.963 |

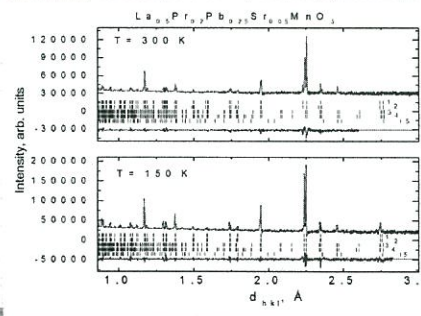


Figure 2. Observed (black), calculated (red) and the difference between the observed and calculated diffractogram (blue) for x=0.05.

In agreement with the obtained neutron data the LPPSMO samples contain two crystalline phases: an rhombohedral one (R-3c) (1) and an orthorhombic phase (Pnma) (3). There are two magnetic phases (2, 4) and a small contribution of hausmanite (5). We attributed the observed difference of the phase composition to a change in chemical composition at the surface comparing with the inner phase composition of the samples.

Magnetic properties We suppose that a part of the Mn^{3+} cations does not participate to the total magnetic moment or they are forming an antiferromagnetic compound, with the same chemical composition and the same structure as the main phase. The substitution of La with Pb, together with the oxygen non-stoichiometry contributes to the appearance of the ferromagnetic interactions, and implicitly to a metallic phase. The variation of magnetization vs. temperature at relatively small intensities magnetic fields is similar to that observed in the samples characterized by the spin glass state.

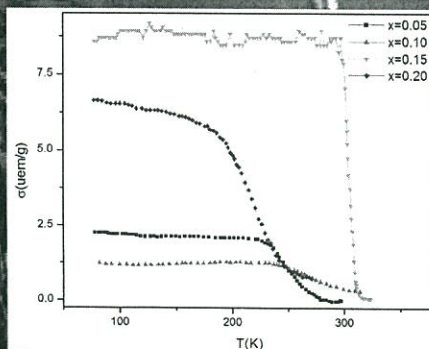


Figure 3. Variation of specific magnetization vs temperature and Pb concentration of LPPSMO manganites, zero magnetic field cooled; measurements were performed at H=0.

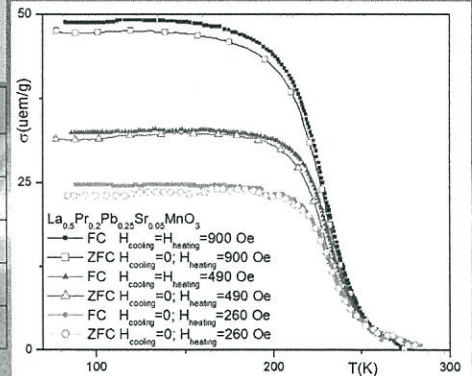


Figure 4. The transition from spin glass to ferromagnetic state for various magnetic field intensities. The spin glass transition was observed at $T > 200$ K (see Fig. 4).

There are two reasons for this behaviour: first, the decrease of the Pb^{4+} concentration and of the oxygen concentration, and second the increase of the Sr^{2+} concentration.

Transport characteristics The transport measurements were performed on field cooled and without field cooled samples, only on heating, with and without applied magnetic field (Fig. 5).

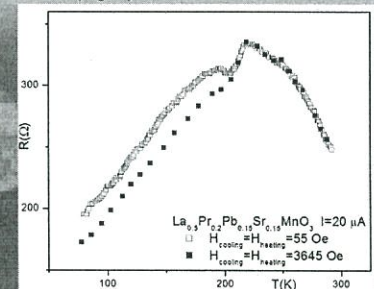


Figure 5. Variation of resistance vs temperature for the applied magnetic field for the sample corresponding to x=0.15

CONCLUSIONS

- We synthesized a set of $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.3-x}\text{Sr}_x\text{MnO}_3$ manganites for various Sr concentrations (x = 0.0, 0.05, 0.10, 0.15 and 0.20). Its contain probably a small amount of hausmanite.
- We have shown that they exhibit a spin glass state to metallic state transition at temperatures between 150 and 250 K.
- At the sample surface a crystallographic transition from a cubic structure to a rhombohedral structure takes place at room temperature when x > 0.05.
- There is a qualitative dependence between the presence of spin glass state and the observed large magnetoresistance

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Influence of spin-glass to ferromagnetic state transition on magnetoresistance in $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.3-x}\text{Sr}_x\text{MnO}_3$ manganites

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The $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.3-x}\text{Sr}_x\text{MnO}_3$ ($x=0.05 - 0.20$) manganites were obtained by ceramic technology, from oxides, mixed in corresponding ratio. X-ray diffraction data at room temperature indicated that the samples contain only a distorted phase: Pm-3m cubic, for $x=0.05$, and R-3c rhombohedral phase (s. Fig. 1) for $x>0.05$. Structure data obtained by neutron diffractometry (ToF method) showed a mixture of two phases: a rhombohedral and an orthorhombic phase. At 150 K neutronograms put in evidence the ferromagnetic phase. Magnetic measurements by VSM method between 77 and 350 K showed that at temperatures lower as 200 K a spin-glass state appeared (see Fig.2).

Curie temperature was determined as a minimum of the first derivative of specific magnetization vs temperature and have a maximum for $x=0.15$, which corresponds to a minimum of the Mn-O distance of rhombohedral phase.

A variation of the concentration of oxygen and its influence on the ratio between $\text{Mn}^{3+}/\text{Mn}^{4+}$ concentrations, implicitly on the magnetic structure of the samples, is discussed.

We put in evidence a strong dependence of the resistance not only on the ratio between Pb and Sr but also of the thermal history of the samples, which can influence the ratio between spin-glass state and ferromagnetic state concentrations at temperature lower as those corresponding transition.

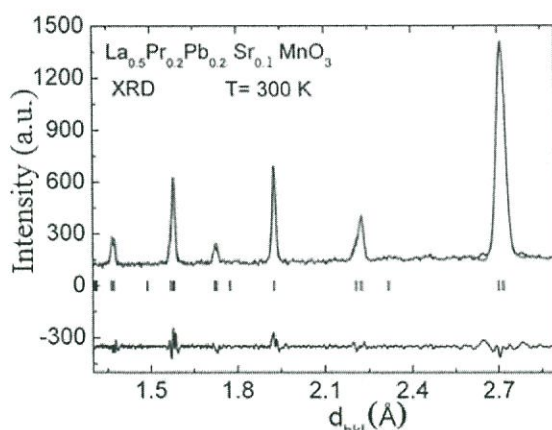


Figure 1 Observed and calculated diffractogram of $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.25}\text{Sr}_{0.05}\text{MnO}_3$; at zero intensity - Bragg maxima; bottom - difference between the observed and calculated diffractograms

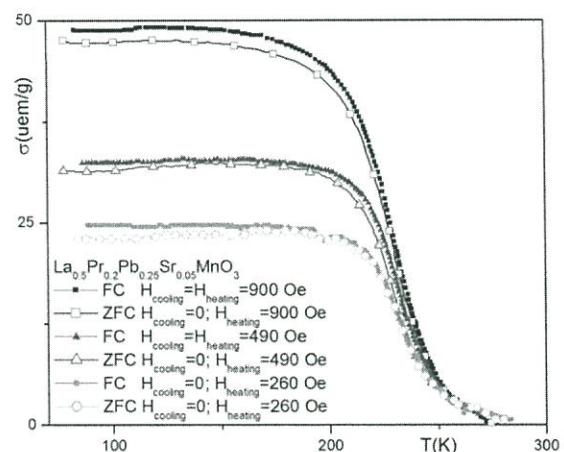


Figure 2 Variation of specific magnetization with temperature for $\text{La}_{0.5}\text{Pr}_{0.2}\text{Pb}_{0.25}\text{Sr}_{0.05}\text{MnO}_3$